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Recent advances in unveiling active sites in molybdenum sulfide-based electrocatalysts for the hydrogen evolution reaction

Bora Seo¹ and Sang Hoon Joo^{1,2*}

Abstract

Hydrogen has received significant attention as a promising future energy carrier due to its high energy density and environmentally friendly nature. In particular, the electrocatalytic generation of hydrogen fuel is highly desirable to replace current fossil fuel-dependent hydrogen production methods. However, to achieve widespread implementation of electrocatalytic hydrogen production technology, the development of highly active and durable electrocatalysts based on Earth-abundant elements is of prime importance. In this context, nanostructured molybdenum sulfides (MoS_x) have received a great deal of attention as promising alternatives to precious metal-based catalysts. In this focus review, we summarize recent efforts towards identification of the active sites in MoS_x -based electrocatalysts for the hydrogen evolution reaction (HER). We also discuss recent synthetic strategies for the engineering of catalyst structures to achieve high active site densities. Finally, we suggest ongoing and future research challenges in the design of advanced MoS_x -based HER electrocatalysts.

Keywords: Hydrogen evolution reaction, Molybdenum sulfide, Electrocatalyst, Active site, Synthetic strategy, Structure engineering, Phase engineering

1 Introduction

Hydrogen is a sustainable and renewable energy carrier that has demonstrated potential as an alternative to fossil fuel energy sources [1, 2]. Currently, hydrogen is produced mainly by steam methane reforming and coal gasification, leading to the ensuing problem of CO₂ release [3, 4]. To provide a more environmentally friendly route to hydrogen power, the development of clean hydrogen production technology is required. Electrocatalytic water splitting represents the most promising solution for producing hydrogen using a carbon-free system [5, 6]. However, the dominant use of Pt-based catalysts for the hydrogen evolution reaction (HER) hinders the widespread implementation of electrocatalytic hydrogen production systems, due to their high costs and limited abundance. Hence, there have been significant efforts to

replace Pt-based catalysts with highly active, durable, and non-precious electrocatalysts for the HER [7–19]. Notable examples of non-precious metal catalysts include metal sulfides, metal phosphides, metal carbides, and heteroatom-doped carbons. Among the various classes of non-precious metal-based electrocatalysts [7-82], nanostructured molybdenum sulfides (MoS_x, x = 2-3) have been most widely studied, owing to their high activities, excellent stabilities, and precious metal-free compositions [7-68]. Although bulk MoS₂ exhibits negligible catalytic activity for the HER [83], pioneering theoretical and experimental works by Nørskov and Chorkendorff have demonstrated that nanostructured MoS_x catalysts are able to catalyze the HER with high efficiency [20, 21]. During the last decade, significant progress has been made in designing MoS_x catalysts at the nanoscale, which has resulted in advanced MoS_x-based catalysts with enhanced HER performances.

In this review, we highlight the key findings reported to date regarding identification of the active sites of

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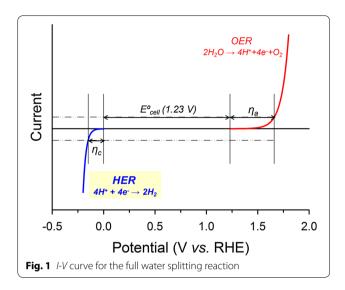
MoS_r catalysts, and synthetic strategies for engineering their structures to yield high active site densities for the HER. Over the past decade, there have been a number of important developments relating to the active sites present in MoS_r catalysts. For example, the edge of MoS₂ was first proposed as a catalytic active site by theoretical calculations in 2005 [20], which was later experimentally demonstrated with a model catalyst composed of MoS₂ nanoparticles grown on a Au(111) surface [21]. Since then, various studies focused on maximizing active edge site densities via structural engineering approaches of MoS_r catalysts, including space-confined growth [22– 25], vertical alignment [26–28], nano-assembly [29–31], and the design of biomimetic molecular catalysts [32-34]. The basal planes of MoS₂, which were believed to be inert in the HER, have also been successfully activated to show meaningful activity by several strategies, including phase engineering from the 2H phase to the metallic 1T phase [26, 35–38], heteroatom doping [39–41], defect site generation [42-47], and strain engineering [40, 48, 49]. Despite significant investigations into the structural engineering of MoS₂-based electrocatalysts to enhance the HER performance, a number of questions remain regarding the active sites and reaction mechanisms. For example, in the case of the amorphous MoS_x, identification of its active sulfur sites for hydrogen adsorption has not yet been clarified due to its structural complexity [50-54].

In this review, we first discuss the basic concepts for the electrocatalytic production of hydrogen, in addition to the activity parameters commonly employed for evaluation of the HER activity. We highlight a number of important results regarding the active sites of MoS_x -based HER catalysts, and summarize representative synthetic strategies for engineering their structures to enhance the number of active sites in different 2H-MoS_2 , 1T-MoS_2 , and amorphous MoS_x structures. We conclude the review by highlighting the current challenges and future research directions in relation to MoS_x -based HER catalysts.

2 Hydrogen evolution reaction

2.1 Hydrogen evolution from water

The electrocatalytic production of hydrogen via water splitting is composed of two half reactions:



In this review, we will focus on the HER taking place at the cathode. Thermodynamically, the HER occurs with 0 V (vs. reversible hydrogen electrode, RHE) of applied potential, as shown in Fig. 1. However, for practical operations, a large excess potential, i.e., the overpotential (η_c), is required, and the development of highly efficient HER electrocatalysts is directly linked to the reduction of the overpotential.

2.2 HER activity parameters

2.2.1 Overpotential to drive a current density of -10 mA cm^{-2}

Generally, the comparison of HER activities has been made in terms of the overpotential at a current density of -10 mA cm^{-2} . This current density corresponds to an efficiency of approximately 10% in solar-to-fuel devices [84], and the derivation is as follows:

- 1. Integration of the solar spectrum (AM1.5G) yields a value of 100 mW cm $^{-2}$, referred to as "1 sun".
- 2. As the redox potential for water oxidation is ~ 1.2 V, an 100% efficient solar-to-fuel device would give 100 (mA V cm⁻²)/(1.2 V) = 83 mA cm⁻² under AM1.5G.
- 3. Thus, a 10% efficient solar-to-fuel device would give $8.3~\mathrm{mA~cm^{-2}}$.

Anode: $2H_2O \leftrightarrow O_2 + 4H^+ + 4e^-$ (oxygen evolution reaction, OER) $E_a = 1.23 \ V - 0.059 \cdot pH \ (V \ vs. \ normal \ hydrogen \ electrode, \ NHE)$ $Cathode: 4H^+ + 4e^- \leftrightarrow \ 2H_2 \ (hydrogen \ evolution \ reaction, \ HER)$ $E_c = 0 \ V - 0.059 \cdot pH \ (V \ vs. \ NHE)$

Therefore, the ranking of HER catalysts by comparison of the overpotentials required to drive a current density of -10 mA cm^{-2} is reasonable in a practical context.

2.2.2 Tafel slope

The Tafel slope is an important kinetic parameter, and is derived from the equation:

$$|\eta| = \frac{2.3RT}{\alpha nF} log \frac{J}{J_0}$$

where η is the overpotential, R is the ideal gas constant, T is the absolute temperature, α is the electrochemical transfer coefficient, n is the number of electrons involved in the electrode reaction, F is the Faraday constant, J is the measured current density, and J_0 is the exchange current density. In addition, J_0 is related to the electrontransfer rate of the reaction, reflecting the intrinsic catalytic activity of the catalyst. From the above equation, the Tafel slope is defined as $\frac{2.3RT}{\alpha nF}$ and bears the units mV dec $^{-1}$. The Tafel slope is therefore determined from the linear regression line of the Tafel plots (η vs. logI), which can be derived from the I-V polarization curve.

The Tafel slope has been used to access the HER mechanism taking place. More specifically, in an acid, the HER proceeds by the initial adsorption of a hydrogen atom on the catalyst surface $(H_{(aq)}^+ + e^- \leftrightarrow H_{ad})$, which is referred to as the Volmer step. Subsequently, molecular hydrogen is produced via the chemical recombination of two H_{ad} atoms $(2H_{ad} \leftrightarrow H_{2~(g)};$ the Tafel step), or through a second electron transfer $((H_{(aq)}^+ + H_{ad} + e^- \leftrightarrow H_{2(g)};$ the Heyrovsky step) [7].

The HER following the Volmer–Tafel mechanism gives rise to Tafel slope of 29 mV dec $^{-1}$, whereas the HER via the Volmer–Heyrovsky mechanism yields 38 mV dec $^{-1}$. In both cases, the combination of two hydrogen atoms is the rate-determining step with lower value indicating faster reaction rate. When the Volmer step is the rate-determining step or the catalyst surface coverage is close to 1, the Tafel slope increases to 116 mV dec $^{-1}$. The most of MoS $_x$ catalysts have shown Tafel slopes in the range of 60–100 mV dec $^{-1}$, following the Volmer–Heyrovsky mechanism.

2.2.3 Turnover frequency

The precise evaluation of each surface site's activity is important to obtain a fundamental understanding of the origin of catalytic activity. The intrinsic activity can be assessed by calculating the turnover frequency (TOF), which is defined as the turnover rate per surface active site. While the comparison of TOFs is meaningful, a fair comparison of TOFs has not yet been carried out due to variations in methods for measuring the active sites in addition to the issues associated with different catalyst structures.

Depending on which sites are assigned as active centers, the TOF can be varied by many orders of magnitude. Theoretically, it is widely accepted that hydrogen atoms bind to surface S sites. However, the majority of studies have calculated TOFs by assuming that surface Mo atoms are the active sites, as the multiple chemical states for S render the calculation difficult. An alternative method is measuring an electrochemically active surface area (ECSA), which can probe all catalytically active surface sites, except the inert areas. Indeed, in some cases, ECSA-derived TOFs have afforded a fairer comparison between different electrocatalysts [80, 82].

It should be noted that the main active sites can differ according to the catalyst structure. For example, only edge sites are active in 2H-MoS_2 , while both edge and basal sites are active in 1T-MoS_2 . For amorphous MoS_x catalysts, the identification of active sites have been difficult due to complexity of their structures. Therefore, a standard method for TOF calculations should be established for fair comparison of MoS_x catalysts with different catalyst structures.

2.2.4 Gibbs free energy

The Gibbs free energy ($\Delta G_{\rm H}$) for atomic hydrogen adsorption has been widely used as an activity descriptor for HER catalysts. According to the Sabatier principle, particularly strong or weak interactions between reaction intermediates and catalysts can lower the overall catalyst efficiency. The HER activity therefore exhibits a volcanoshaped relationship as a function of $\Delta G_{\rm H}$ [21]. Among the various catalysts examined, Pt catalyst exhibits the highest HER performance, with a $\Delta G_{\rm H}$ value close to zero. It has been employed as a figure-of-merit for examining the HER performances of newly developed electrocatalysts. Generally, $\Delta G_{\rm H}$ values are deduced by theoretical calculations using a simplified model to reflect the experimental conditions.

3 Active sites in molybdenum sulfides

3.1 Active edge sites

The active edge sites of MoS_x -based HER electrocatalysts bear a remarkable resemblance to those of the nitrogenase and hydrogenase enzymes, which exhibit excellent activities and selectivities in their natural systems (Fig. 2a) [20]. To date, considerable progress has been made to synthesize highly active and stable heterogeneous electrocatalysts whose structures mimic the active sites of such natural catalysts [32–34]. In 2005, theoretical calculations by Nørskov and coworkers revealed that the ΔG_{H} value of the MoS_2 edge is almost thermo-neutral, suggesting that the MoS_2 edge is a highly plausible active site for the HER (Fig. 2b) [20]. Indeed, this was demonstrated experimentally using a

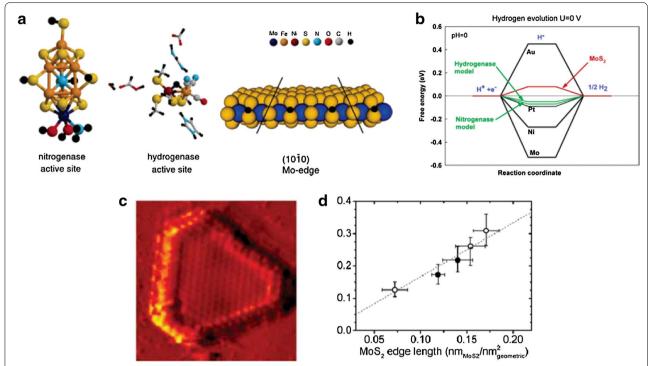


Fig. 2 a Active sites of the nitrogenase and hydrogenase, and depiction of the Mo-edge on MoS_2 slab. **b** Free energy diagram for hydrogen adsorption. **c** STM image of MoS_2 nanoparticles on a Au(111) surface. **d** The exchange current density versus the MoS_2 edge length (Figures reprinted with permission from Refs. [20, 21])

model catalyst comprising MoS_2 nanoparticles grown on a Au(111) surface (Fig. 2c) [21]. This work revealed that the electrochemical HER activity exhibits a linear correlation with edge length of MoS_2 (Fig. 2d). These pioneering works have provided an important evidence that the active sites of MoS_2 are located at the edge planes. Since these findings were reported, significant advances have been achieved in increasing the density of active edge sites [22–34], as will be discussed in Sect. 4.1.

Depending on the size of the MoS_2 nanosheets employed, the edges of MoS_2 can be covered with 0, 50, 75, or 100% sulfur atoms [55, 85, 86], where the sulfur coverage on the edges can significantly affect the adsorption of H atoms, which is directly correlated to HER kinetics. DFT calculations revealed that the most favorable edge configurations correspond to Mo edges covered by 50% S, where $\Delta G_H = 0.06$ eV [87]. In addition to the sulfur coverage, the hydrogen coverage on the edges is also an important factor to determine the hydrogen binding energy [88].

3.2 Active sites in 2H- and 1T-MoS₂

 ${
m MoS}_2$ has several polymorphs with distinct atomic configurations and electronic structures. Among these

polymorphs, 2H-MoS₂ and 1T-MoS₂ structures are the most widely investigated for use as electrocatalysts in the HER. These polymorphs exhibit trigonal prismatic and octahedral unit cell structures, respectively (Fig. 3a-c) [17]. In addition, 1T-MoS₂ has a dense atomic configuration in the basal surfaces and a high electronic conductivity, which is six orders of magnitude greater than that of 2H-MoS₂, thereby resulting in an enhanced HER performance of 1T-MoS₂ compared to 2H-MoS₂. However, the preparation of 1T-MoS2 is more complicated than that of 2H-MoS₂, due to its metastable nature. In this regard, significant efforts have been directed to prepare the stable and highly pure 1T-MoS₂ polymorph using chemical Li-intercalation [35, 36], electrochemical Li-intercalation [26, 37], and a pressurized hydrothermal process [38]. For example, The Chhowalla group prepared the 1T phase MoS₂ through an exfoliation reaction using lithium borohydride, where a significantly higher yield was obtained than that using n-butyllithium (i.e., 80% vs. ~50%) [36]. Using the exfoliated 1T-MoS₂ sample, the main active sites of the two polymorphs, i.e., 2H-MoS₂ and 1T-MoS₂, were compared. Upon partial oxidation of the edges, the HER activity of 2H-MoS₂ was significantly reduced, while that of 1T-MoS₂ remained unaffected, suggesting that the main active site of the 2H-MoS₂ polymorph is located at

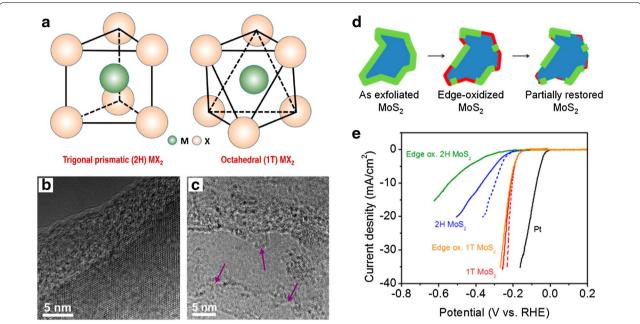


Fig. 3 a Unit cell structures of 2H-MoS₂ and 1T-MoS₂. HRTEM images for **b** 2H-MoS₂ and **c** 1T-MoS₂. **d** Schematic representation of the oxidation process and partial restoration of the MoS₂ edges after several voltammetric cycles. **e** HER polarization curves of 1T and 2H MoS₂ before and after edge oxidation. *Dashed lines* indicate the iR-corrected polarization curves (Figures reprinted with permission from Refs. [17, 35, 36])

the edge sites, while that of 1T MoS₂ may be located at basal sites. (Fig. 3d, e).

3.3 Activating inert basal sites

The basal plane of the 2H-MoS₂ polymorph has long been considered inert towards the HER, essentially rendering the large basal surface useless. However, several theoretical calculations have suggested that the inert basal planes of MoS₂ can be exploited as potential active sites following activation by heteroatom doping, defect site generation, and strain engineering (Fig. 4) [39-49]. For example, heteroatom doping into MoS₂ can produce high dopant concentrations on the surface, thereby modifying the hydrogen absorption strength of nearby surface atoms (Fig. 4a). More specifically, Du and co-workers suggested that doping MoS2 with a heteroatom in combination with a small compressive strain can yield an ideal $\Delta G_{\rm H}$ for hydrogen binding in the HER (Fig. 4b, c) [40]. In addition, defects are known to perturb the local density of states, creating additional energy levels below the conduction bands [89]. In this context, Wang and co-workers evaluated the effect of sixteen different structural defects on activating the basal plane of MoS2 monolayers (Fig. 4d) [43]. The theoretical results suggested that six defects, including sulfur vacancies, greatly enhanced the HER performance of MoS₂ (Fig. 4e). These theoretical findings were later verified experimentally, as discussed in Sects. 4.3 and 4.4.

3.4 Active sites in amorphous MoS_x

In addition to the crystalline MoS₂ structure, amorphous MoS_r has attracted significant attention due to its facile preparation under mild conditions, such as wet chemical synthesis [60] and electrodeposition [58, 59, 61]. Unlike MoS₂, the active sites present in amorphous MoS_r have received little attention due to the complex polymeric structure of such compounds (Fig. 5a) [52]. It contains short-range atomic arrangements with [Mo₃S₁₃]²⁻ clusters as building units (Fig. 5b). Similar to MoS₂, the question of active sites on amorphous MoS_x has been unavoidably and continuously raised. Recently, the sulfur atoms present in amorphous MoS_r have been directly confirmed as the catalytic active sites for the HER via operando Raman spectroscopic analysis [54]. However, the sulfur atoms exist in four key states, namely bridging S_2^{2-} , terminal S_2^{2-} , unsaturated S^{2-} , and apical S^{2-} . Due to the diverse sulfur chemical states present in the amorphous MoS_r, identification of the catalytically active sulfur sites for proton reduction is challenging. Interestingly, Yeo and co-workers reported a linear correlation between TOFs for the HER and the percentage of S species with higher electron binding energies using X-ray photoelectron spectroscopy (XPS) (Fig. 5c) [53]. This work suggested bridging ${\rm S_2}^{2-}$ species as the potential catalytic active sites. In addition, Yano and Hu and co-workers investigated the structural changes taking place in the amorphous MoS_x under HER conditions

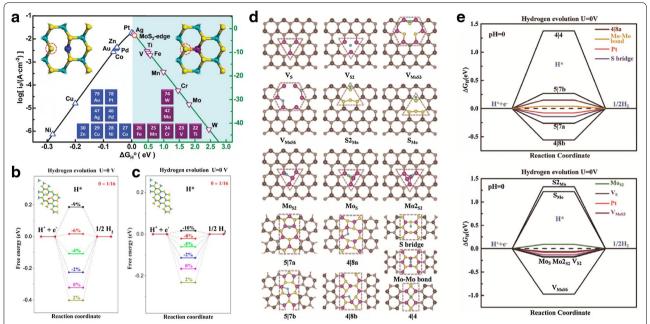


Fig. 4 a The volcano-shaped relationship between the ($\log(i_0)$) and ΔG_H . Free energy diagram for the HER on **b** Rh-doped MoS₂, and **c** Ag-doped MoS₂, at different strains. **d** 16 Examples of different structural defects. **e** ΔG_H of the HER process on each defect region (Figures reprinted with permission from Refs. [39, 40, 43])

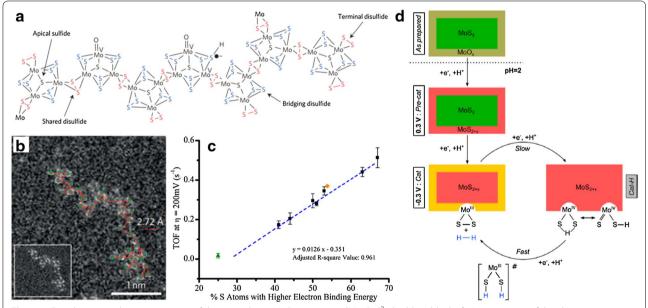


Fig. 5 a Coordination polymeric structure of the amorphous MoS_x containing $[Mo_3S_{13}]^{2-}$ building blocks. **b** Arrangement of the cluster units in a 1D unfolding chain. **c** TOFs versus the percentage of S atoms with high electron binding energy. **d** Proposed catalytic cycle for the HER over amorphous MoS_x (Figures reprinted with permission from Refs. [51–53])

using in situ X-ray absorption spectroscopy (Fig. 5d) [51]. They proposed a reaction mechanism, where the catalytic species is similar to MoS₂, which corroborates an earlier result by Nilsson and Jaramillo and co-workers

[50]. Although significant efforts have been devoted to revealing the active sites for proton reduction, the identification and confirmation of genuine catalytically active sulfur sites remain elusive.

4 Synthetic strategies for increasing active site densities

4.1 Nanospace-confined growth

Reducing the particle size of MoS₂ is the most straightforward method that can increase the density of active edge sites. However, thermodynamics tends to favor growth through the basal plane because the formation of edge sites is highly energetic due to the undercoordinated atomic configuration. To overcome this challenge, confinement growth within a nanospace has been reported [22–25]. The Jaramillo group successfully synthesized a mesoporous MoS₂ structure using a silica template with double-gyroid (DG) morphology (Fig. 6a) [22]. The resulting DG MoS₂ structure exhibited a high surface curvature, thereby exposing a large fraction of active edge sites. The DG MoS₂ exhibited a higher HER

performance than high aspect-ratio core-shell MoO₃-MoS₂ nanowires (Fig. 6b). In addition, the DG MoS₂ gives a Tafel slope of 50 mV dec⁻¹, which is relatively low compared to previously reported MoS2-based HER catalysts (Fig. 6c). In this direction, our group prepared layer number-controlled MoS₂ nanosheets using the nanospace confined-growth approach [23]. In this work, the pores of mesoporous silica templates were partially filled with carbon, and MoS₂ structures were subsequently grown inside the residual nanospace of the silica-carbon composites. After the etching of silica template, MoS₂ nanostructures embedded in the frameworks of ordered mesoporous carbons (MoS₂@OMC) were generated. As shown in the TEM images (Fig. 6d), the formation of an extended basal plane was successfully hindered with a size <5 nm in the lateral direction. It was found that

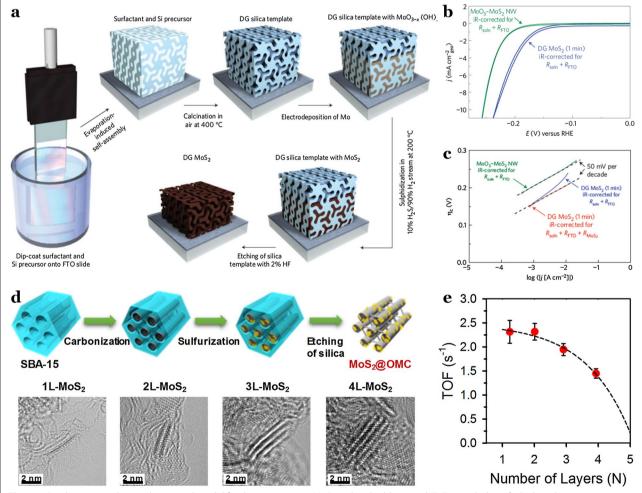


Fig. 6 a Synthetic procedure and structural model for the mesoporous MoS_2 with a double-gyroid (DG) morphology. **b** Cyclic voltammogram of DG MoS_2 (1 min sample) versus core–shell MoO_3 - MoS_2 nanowires (NW) at 5 mV s⁻¹. **c** Tafel plot of DG MoS_2 (1 min sample) versus core–shell MoO_3 - MoS_2 NW. **d** Synthetic procedure for the MoS_2 nanosheets embedded on ordered mesoporous carbon nanorods ($MoS_2@OMC$), and TEM images for 1L-, 2L-, 3L-, and 4L- $MoS_2@OMC$. **e** TOFs versus the number of layers in the MoS_2 (Figures reprinted with permission from Refs. [22, 23])

the TOF increased upon decreasing the layer number in MoS_2 , and this trend in activity could be correlated to the physical and chemical properties of MoS_2 nanoplates (Fig. 6e). These results indicate that space confinement growth paves the way to controlling the surface structure and size of MoS_2 at the nanoscale to ultimately develop effective catalysts with high densities of active edge sites at the surface.

4.2 Phase engineering

As mentioned in Sect. 3.2, the active sites, which were limited to the edges in 2H-MoS2, have been expanded to basal surfaces via phase engineering from trigonal prismatic (2H-MoS₂) to metallic octahedral (1T-MoS₂) structures (Fig. 7a, b). For example, the Jin group demonstrated that the chemical exfoliation of 2H-MoS₂ using n-butyllithium significantly enhances the HER activity through the formation of metallic 1T-MoS₂ [35]. This phenomenon was ascribed to the fast electron transport and increased number of active sites guaranteed by the metallic 1T-MoS₂ nanosheets. These 1T-MoS₂ nanosheets required a low overpotential of 187 mV to drive a current density of -10 mA cm^{-2} , compared to ~313 mV for 2H-MoS₂ (Fig. 7c). Furthermore, the Tafel slope of 43 mV dec⁻¹ for 1T-MoS₂ was significantly lower than that of $2H\text{-MoS}_2$ (i.e., 110 mV dec⁻¹), indicating fast HER kinetics in 1T-MoS2 (Fig. 7d). As an alternative phase engineering technique, the Cui group reported the use of an electrochemical Li intercalation method to generate the 1T-MoS₂ phase (Fig. 7e) [26]. This method allowed the vertically-aligned 2H-MoS₂ to be converted into 1T-MoS₂, which exhibited an enhanced HER performance (Fig. 7f, g). In addition to the intercalation of Li ions, mechanical strains also induced the partial formation of 1T-MoS₂ structures, thereby activating the HER [67]. However, despite numerous reports focusing on the use of phase engineering to enhance HER performances, the origin of the HER activity has not yet been completely elucidated. Decoupling of the intrinsic activities of 1T phase from the overall HER activity is required to reach a fundamental understanding of the active sites present in MoS₂ polymorphs.

4.3 Heteroatom-doping

The incorporation of heteroatoms into the basal surface of MoS₂ nanosheets can significantly modify the electronic structure of in-plane S atoms neighboring the heteroatom, thereby altering the adsorption strength of H atoms. In this context, the Bao group reported the doping of single Pt atoms into the in-plane domain of MoS₂ nanosheets (Pt-MoS₂) [39], where the resulting Pt-MoS₂ exhibited an enhanced HER performance compared with the undoped MoS_2 . Furthermore, they also screened the HER activities of MoS₂ doped with a number of transition metals, resulting in volcano-shaped relationships with the adsorption free energy of the H atoms ($\Delta G_{\rm H}$) (Fig. 4a). Their study suggests a novel method for activating the inert in-plane domain of MoS2 catalysts, which may also be extended to other 2D materials applicable in a variety of catalytic reactions.

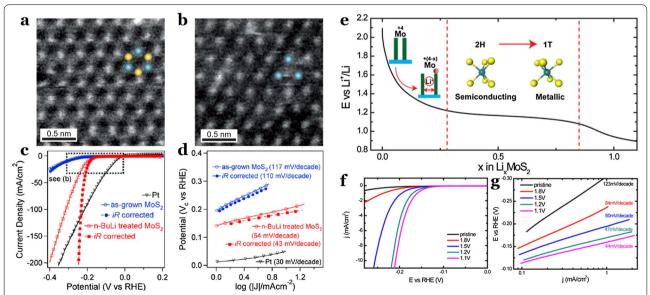


Fig. 7 Dark-field scanning transmission electron microscopy images of **a** 2H-MoS₂, **and b** 1T-MoS₂. **c** Polarization curves of the 2H- and 1T-MoS₂ for the HER, and **d** the corresponding Tafel plots. *Filled symbols* indicate the iR-corrected data. **e** Galvanostatic discharge curve representing the lithiation process. **f** Polarization curves of the pristine and lithiated MoS₂ for the HER, and **g** the corresponding Tafel plots (Figures reprinted with permission from Refs. [9, 26, 35])

4.4 Defect and strain engineering

The inert basal surfaces of 2H-MoS2 have also been successfully activated by creating defect sites and/or inducing strain [42-47]. The first example of defect engineering conducted by Xie and co-workers focused on the exposure of additional active edge planes by forming cracks on the surfaces of nanosheets (Fig. 8a) [42]. They reported that defect-rich MoS₂ exhibited a significantly enhanced HER performance compared with defect-free MoS₂ (Fig. 8b). In addition, the Ajayan group demonstrated that oxygen plasma treatment and H2 annealing introduced additional active sites within the MoS2 monolayer, significantly improving the HER activity [44]. Recently, more rational and controllable defect modulation has been reported through combined experimental and theoretical studies [45-47]. Allwood and co-workers prepared MoS₂ nanocrystals and activated the Mo atoms in the basal surface of MoS₂ nanocrystals by S depletion [45], with the resulting activated MoS₂ exhibiting an very high HER performance (\sim 150 mV at -10 mA cm⁻² and a Tafel slope of \sim 29 mV dec $^{-1}$). Cao and co-workers also verified the importance of S vacancies on the catalytic activity for the HER [46], estimating the intrinsic TOFs of the edge sites, S vacancies, and grain boundaries as approximately 7.5, 3.2, and 0.1 s⁻¹, respectively. Finally, the Zheng and Nørskov groups reported that straining of the S-vacancies further enhances the HER activity (Fig. 8c-e) [48]. The experimental results was further verified with theoretical results that optimum level of strain and S-vacancy can tune the $\Delta G_{\rm H}$ close to zero, guaranteeing the highest intrinsic HER activity.

5 Summary and future challenges

In this review, we have highlighted recent major achievements regarding elucidation of the active sites in MoSxbased electrocatalysts for the HER, and summarized synthetic strategies for designing MoS_x electrocatalysts with enhanced HER activities. The edge site of MoS₂ was initially identified as the active site for the HER, which then triggered the development of new HER catalysts that can maximize the density of active edge sites, thereby boosting HER activity. The basal surface of MoS₂, which were previously believed to be inert towards the HER, can be converted into HER active species via appropriate structural engineering, which include phase transformation, heteroatom doping, defect site generation, and strain engineering. In addition to the crystalline MoS_2 , amorphous MoS_x has also been extensively studied as an efficient electrocatalyst for the HER. Amorphous MoSx can possess abundant active edge structures originating from the building blocks of [Mo₃S₁₃]²⁻ clusters, however, the multiple chemical states of sulfur in such species hamper identification of the actual active sulfur states. A comprehensive and systematic study to reveal the key active sites in different MoSx structures still remains a challenging task. Understanding of active

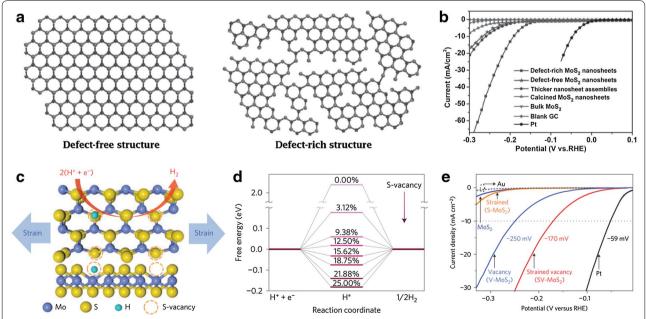


Fig. 8 a Structural models of the defect-free and defect-rich MoS_2 structures. **b** Polarization curves of the defect-free and defect-rich MoS_2 structures for the HER. **c** Schematic representations of the *top* and *side* views of MoS_2 containing strained S-vacancies on the basal planes. **d** Free energy versus the HER reaction coordinate for the S-vacancy range of 0–25%. **e** Polarization curves for the strained, vacancy, and strained vacancy MoS_2 for the HER (Figures reprinted with permission from Refs. [42, 48])

sites would enable high-performance MoSx electrocatalysts for the HER.

Authors' contributions

BS and SHJ wrote the manuscript. Both authors read and approved the final manuscript.

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Acknowledgements

Not applicable.

Competing interests

The authors declare that they have no competing interests.

Funding

This work was supported by the Korea Institute for Advancement of Technology (KIAT) funded by the Ministry of Trade, Industry and Energy (MOTIE) (KIAT_N0001754) and the Korea Evaluation Institute of Industrial Technology (KEIT) funded by the MOTIE (10050509). B.S. acknowledges the Global Ph.D. Fellowship (NRF-2013H1A2A1032647).

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Received: 10 June 2017 Accepted: 7 July 2017 Published online: 25 July 2017

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